

In Situ Growth of Epitaxial YAlO_3 Thin Films by Metal-Organic Chemical Vapor Deposition

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The successful integration of high-temperature superconducting (HTS) materials into active and passive microelectronics technologies depends crucially upon advances in the fundamental science of thin film formation and performance.¹ To this end, considerable progress has recently been made in the growth of HTS films by both physical vapor deposition (PVD) and metal-organic chemical vapor deposition (MOCVD) techniques.¹ Equally important for multilayer device fabrication is the quest for lattice-matched, chemically compatible, low dielectric constant/low dielectric loss materials for use as substrates, buffers, dielectrics, insulators, and overlayers.²⁻⁴ As in HTS film formation, MOCVD offers the attraction for insulating ceramic film growth of the ability to coat complex shapes, simplified apparatus, adaptability to large scale/area depositions, and depositions at low temperatures.⁵⁻⁷ YAlO_3 is an example of a promising insulating material for HTS device applications, having

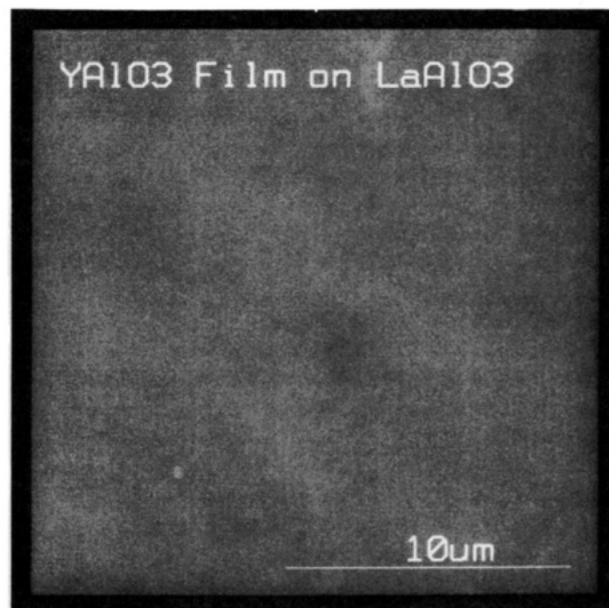


Figure 1. Scanning electron micrograph of an as-deposited MOCVD-derived YAlO_3 film on a (110) LaAlO_3 substrate.

a good lattice and thermal expansion match with YBCO, BSCCO, and TBCCO as well as excellent dielectric properties ($\epsilon \approx 16$ at 77 K, 10 GHz; $\tan \delta = 1 \times 10^{-5}$ at 77 K, 10 GHz).⁸ Furthermore, YAlO_3 exhibits no phase transitions between 25 and 1300 °C, which in other materials results in twinning and degradation of film properties.⁸ We report here the first in situ (not requiring a postanneal) deposition of phase-pure epitaxial thin films of YAlO_3 by MOCVD.

Efficient MOCVD processes depend crucially upon the development of high-purity metal-organic precursors with a high and stable vapor pressure as well as predictable gas-phase reactivity. Strategies for precursor design include encapsulation of the metal ions in sterically encumbered, nonpolar ligation environments.^{9,10} In the present case, the complexes $\text{Y}(\text{dpm})_3$ (dpm = dipivaloylmethanate), $\text{Al}(\text{acac})_3$ (acac = acetylacetonate) represent an embodiment of this strategy and are successfully employed as volatile metal-organic precursors for YAlO_3 .

MOCVD of YAlO_3 films was carried out in a horizontal stainless steel reactor having individual inlet tubes for introducing the volatile, metal-organic precursors as well as the reactant gas N_2O . Deposition was carried out at a system pressure of 1.5 Torr (background pressure = 0.10 Torr), and the single crystal (110) LaAlO_3 substrate (indexed here in the rhombohedral system) was heated resistively to 800 °C. N_2O (99.0%) was introduced at 150 sccm immediately upstream of the susceptor. The precursors $\text{Y}(\text{dpm})_3$ and $\text{Al}(\text{acac})_3$ were synthesized¹¹ from

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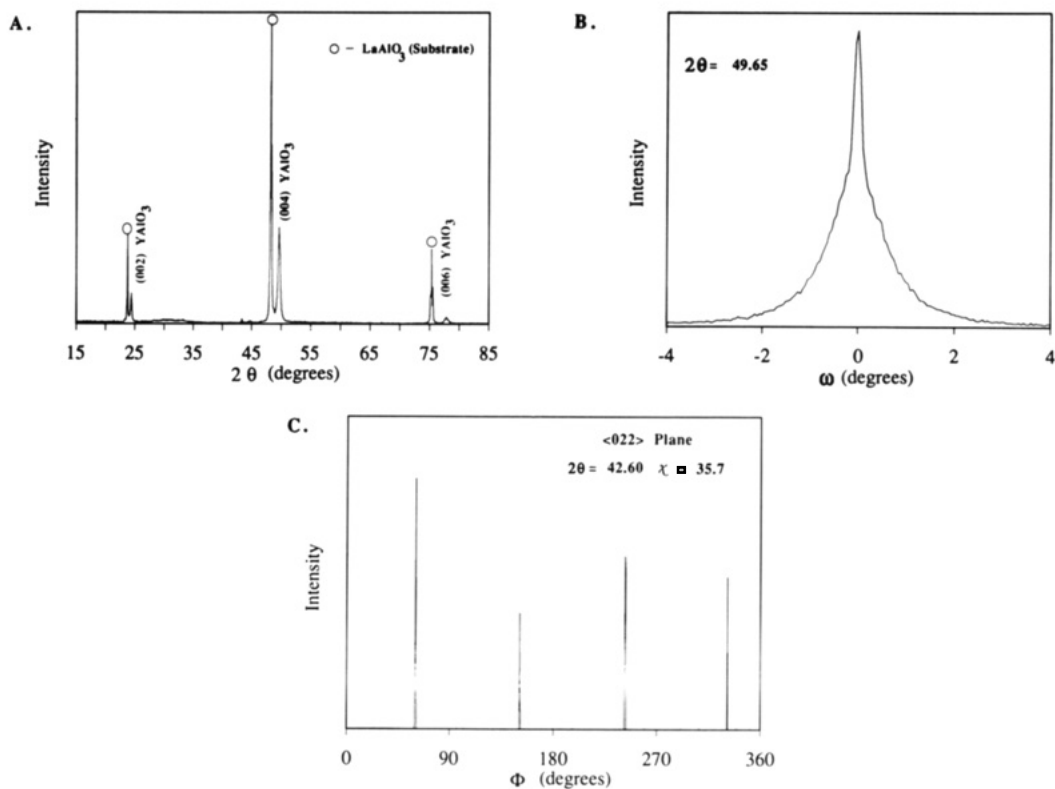


Figure 2. (A) θ - 2θ X-ray diffraction scan of a YAlO_3 film deposited by in situ MOCVD on a (110) LaAlO_3 substrate. (B) X-ray diffraction ω scan rocking curve of the (004) reflection of an MOCVD-derived YAlO_3 film on a (110) LaAlO_3 substrate. The fwhm of the reflection is 0.44° . (C) X-ray diffraction ϕ scan of an MOCVD-derived YAlO_3 film on a (110) LaAlO_3 substrate.

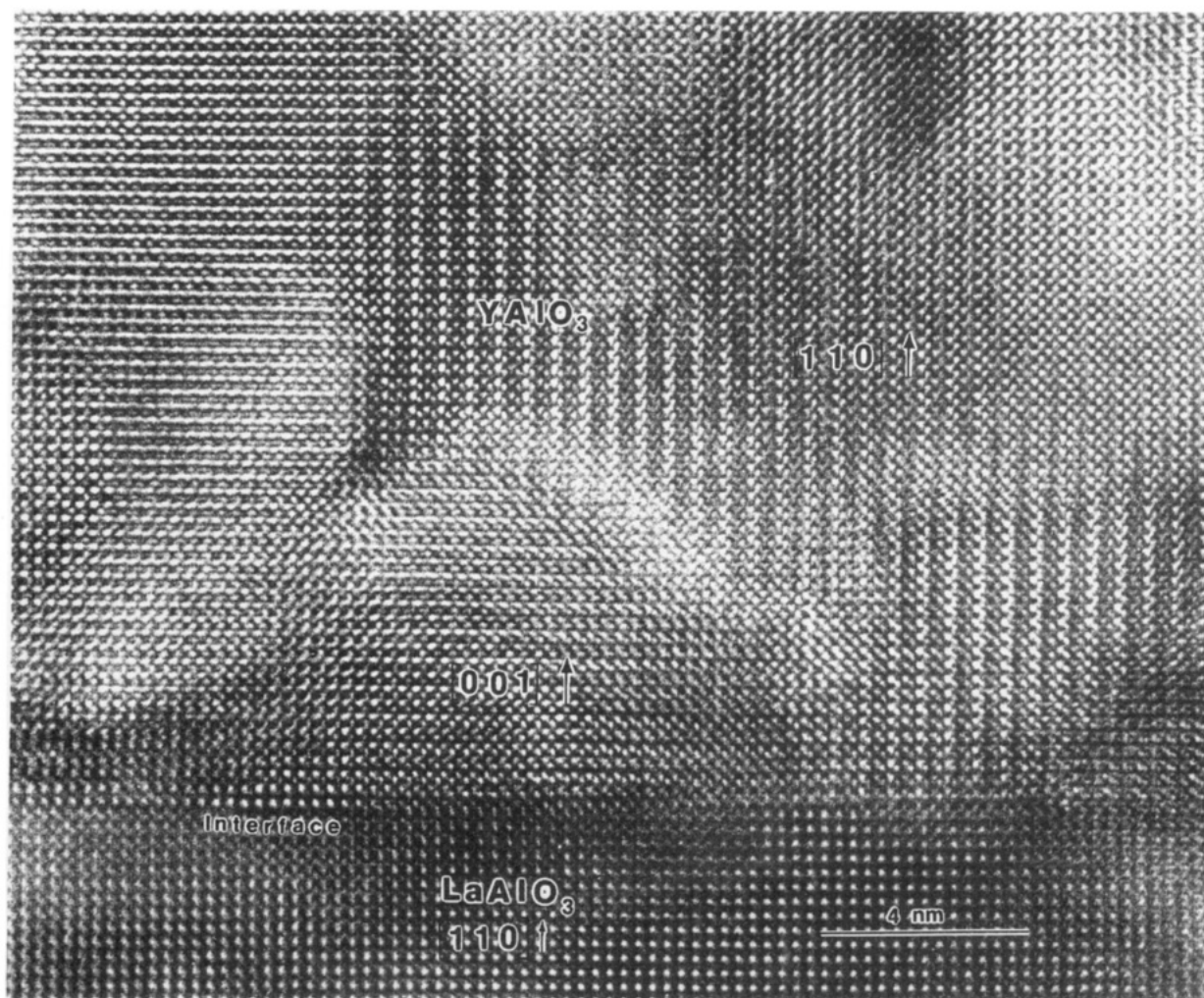


Figure 3. Cross-sectional HREM image of an MOCVD-derived YAlO_3 film on (110) LaAlO_3 .

high-purity metal starting materials and were multiply sublimed. The precursors (~ 5.0 g each) were contained in individual thermostated (100°C) 250-mL Pyrex round bottom flask reservoirs and were transported to the reaction chamber by Ar flowing at 100 and 40 sccm, respectively. Typical film growth rates were $\sim 1.0\ \mu\text{m/h}$.

As evidenced by SEM images (Figure 1), the present in situ grown YAlO_3 films have a smooth, mirrorlike surface, in contrast to typical results for metal oxide films grown by two-step MOCVD/postanneal processes.¹² X-ray diffraction (Ni-filtered $\text{Cu K}\alpha$ radiation) θ - 2θ scans of the MOCVD-derived YAlO_3 films (Figure 2A) reveal phase purity and a high degree of texturing. The coincidence of the (002) and (110) diffraction planes renders initial differentiation of film (001) and (110) growth orientations ambiguous (*vide infra*). Diffractometric rocking curves (ω scans) performed with a double crystal diffractometer ($\text{Cu K}\alpha$ radiation) indicate a high degree of alignment/perfection of the film growth planes with respect to the substrate surface. Thus, the full width at half maximum (fwhm) of the YAlO_3 (004)/(220) reflection was determined by least-squares fitting of the data to be 0.44° (Figure 2B) versus 0.11° for the corresponding (220) reflection of the single-crystal substrate. A ϕ scan was also performed to assess the quality of the in-plane epitaxy using a diffractometer ($\text{Cu K}\alpha$ radiation) equipped with a four-circle goniometer. In theory, four equivalent planes of reflection should be observed, repeating every 89° or 91° . A typical ϕ scan of the $\langle 202 \rangle$ family of diffraction planes (Figure 2C) exhibits the requisite 2-fold (nearly 4-fold) symmetry, hence demonstrating a high level of in-plane epitaxy.

Cross-sectional high-resolution electron microscopy (HREM) was used to further define the microstructure of the epitaxial YAlO_3 growth. As can be seen in Figure 3, the films show sizeable regions of single-crystal, defect-free, and epitaxial growth with atomically abrupt film-substrate interfaces. The HREM images and selected area diffraction patterns (Figure 4) also reveal that the present MOCVD-derived YAlO_3 films actually consist of two types of structurally related domains which are not readily distinguishable by standard X-ray diffraction techniques: one with the c axis perpendicular to the substrate surface and the other with the $[110]$ direction perpendicular to the substrate surface, corresponding to growth with (001) and (110) orientations, respectively.

In summary, these results demonstrate that phase-pure, highly oriented, epitaxial thin films of the perovskite

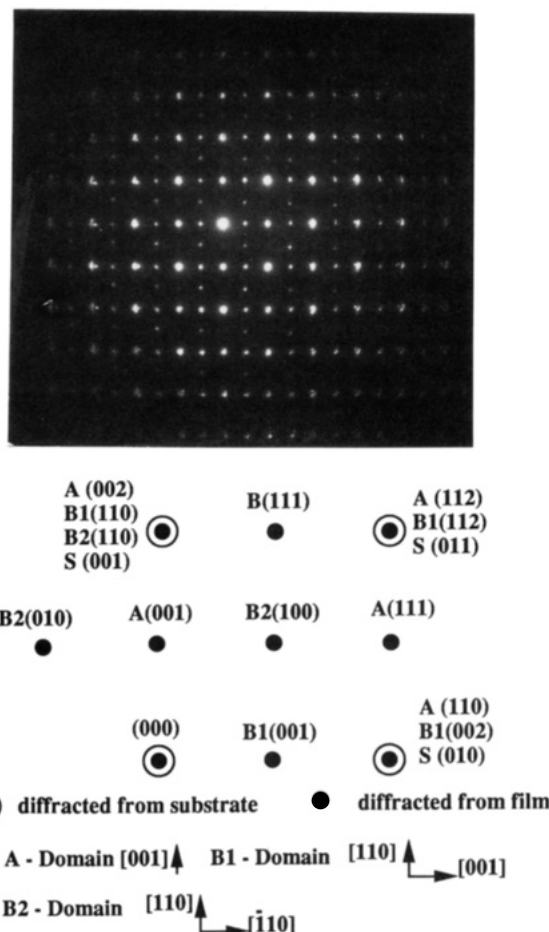


Figure 4. Selected-area diffraction pattern of an MOCVD-derived YAlO_3 film on (110) LaAlO_3 by TEM. The electron beam is perpendicular to the film surface.

dielectric YAlO_3 can be efficiently grown in situ on (110) LaAlO_3 substrates by MOCVD at 800°C using Y and Al β -diketonate precursors. Application of this approach to the fabrication of more elaborate multilayer architectures as well as extensions to other dielectric and photonic metal oxide materials is currently under investigation.

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